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Spin-manipulated phonon dynamics during magnetic phase transitions in triangular lattice antiferromagnet $CuCr_{1-x}Mg_xO_2$ semiconductor films

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Raman scattering and infrared reflectance spectra of $CuCr_{1-x}Mg_xO_2$ films ($x = 0.03$, 0.06 and 0.09) in the temperature range of 5–300 K have been studied, combined with first-principles calculations. The abnormal redshift of the E_g Raman mode center with decreasing temperature below 100 K for the lightly doped film ($x = 0.03$) is ascribed to the Cr 3d–O 2p–Cu 3d interaction. Strong disturbance of the local spin fluctuation at Cr sites in heavily Mg-doped films ($x = 0.06$ and 0.09) drives the E_g mode center to a normal blueshift with decreasing temperature. With further decreasing temperature, the out-of-plane structure increases the internal spin-charge coupling. It enhances the spin-flip splitting and brings back the abnormal redshift of the E_g Raman mode center. A similar but more obvious trend can be found from temperature-dependent E_u infrared mode center shifts. Two successive magnetic transitions were observed at the corresponding Néel temperatures T_{N1} around 24.7 K and T_{N2} around 23.0 K, as manifested by magnetoresistance measurements. The interesting phenomena of phonon dynamics are suggested to be manipulated by the spin structures during the magnetic transitions.

1 Introduction

Geometrically frustrated magnetic systems have been paid much attention in recent years owing to the emergence of their prominent magnetic properties.[1,2] A large degeneracy of the ground state of the system can be yielded in frustrated magnetic systems, turning the matter into a fluid-like or wave-like state. This is because the localized magnetic moments or spins interacting through competing exchange interactions cannot be simultaneously satisfied.[3] The Heisenberg triangular-lattice antiferromagnet (HTAF) is an excellent model for low-dimensional and frustrated magnetism, although some factors, like additional spin-anisotropy terms, usually make these systems different from the perfect one.[4] As one of the most

typical frustrated spin systems, delafossite-structured $CuCrO_2$ has attracted considerable attention due to various favorable properties. High transmittance and low resistivity make impurity-doped $CuCrO_2$ a potential candidate for p-type transparent conducting oxide (TCO) semiconductors.[5–8] Meanwhile, the magnetoelectric properties of the $CuCrO_2$ system have also attracted a lot of attention, due to the peculiar spin characteristics aroused by the Cr^{3+} ions. Consequently, how the spin property of the $CuCrO_2$ system evolves under external fields has been a research hotspot, since the equilibrium spin structures are expected to be sensitively dependent on both thermal and magnetic fluctuations.[8,9]

More recently, it has been reported that $CuCrO_2$ undergoes two successive magnetic transitions in the low temperature regions.[10] On top of this revelation, a suggestion has been proposed that the triangular lattice antiferromagnet (TLA) structure of $CuCrO_2$ can provide an opportunity for unique magnetoelectric control of spin-chiral ferroelectric domain structures by means of both electric or magnetic fields.[11,12] It opens up promising applications for the realization of future devices based on $CuCrO_2$. As advanced techniques that study shifts in energy with information about phonon vibrational modes in a system, Raman scattering and infrared spectra are commonly used to study the intrinsic electromagnetic properties and crystal structures of a system. Nevertheless, few studies have been carried out to expound the influence of spin structure on phonon dynamics for $CuCrO_2$ systems. The intrinsic correlation between optical properties and magnetic transitions for $CuCrO_2$ systems has not been reported up to now. Moreover, it has been suggested that Mg-doped $CuCrO_2$ has some advantages over pure $CuCrO_2$, which has much lower conductivity. Upon doping with 5% Mg, the conductivity can be increased to 220 S cm^{-1}, which is the highest among the delafossite-structured ABO_2 systems.[5] This property makes them easier to combine with their traditional n-type TCO counterparts. Neutron diffraction measurements have been studied to explore the magnetic properties of Mg-doped $CuCrO_2$. A three-dimensional (3D) long-range character of the magnetic order

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was proposed to fit the neutron diffraction results instead of the traditional two-dimensional (2D) HTAF model.[13] However, the influence on the spin characteristics of $CuCrO_2$ from Mg doping has not been discussed in detail because only $CuCr_{0.98}Mg_{0.02}O_2$ was studied in those measurements. This limits the development of spintronic devices with delafossite-structured $CuCrO_2$ materials.

In this article, the Raman and infrared spectra of $CuCr_{1-x}Mg_xO_2$ ($x = 0.03$, 0.06 and 0.09) films have been measured in the temperature region of 5–300 K. The lattice vibration properties have been studied by investigating the experimental observations and first-principles calculations. The magnetoresistance (MR) effect has also been investigated under different magnetic fields to offer more information about the spin structures. By analyzing the optical properties and MR effect, one can discover the intrinsic mechanism of the spin characteristics, which is important for the development of spin transportation for $CuCrO_2$-based devices.

2 Experimental and calculation details

$CuCr_{1-x}Mg_xO_2$ ($x = 0.03$, 0.06 and 0.09) films were prepared on (001) sapphire substrates by the sol–gel route, employing a spin-coating process. A detailed fabrication procedure and the film structure parameters can be found in a previous report.[14] Temperature-dependent Raman scattering measurements were carried out using a Jobin-Yvon LabRAM HR 800 UV micro-Raman spectrometer. A He–Ne laser with a wavelength of 632.8 nm is taken as the excitation source. The sample was mounted into a continuous flow liquid helium cryostat (Oxford cryostat MicrostatHiResII system) in the temperature range of 5 K to 300 K with a resolution of 0.1 K. To get rid of the trivial temperature dependence, all Raman spectra have been divided by the Bose–Einstein occupation number $n(\omega) + 1 = 1/[1 - \exp(-\hbar\omega/k_B T)]$ ($\hbar$ and k_B are the Planck constant and Boltzmann constant, respectively). Note that the derived Raman spectra can be fitted with independent damped harmonic oscillators to uniquely obtain the phonon frequency.

Infrared reflectance spectra were recorded using a Bruker Vertex 80V Fourier transform infrared (FT-IR) spectrometer equipped with a specular reflectance setup. The spectra were measured with a Globar lamp (a U-shaped SiC piece), a 6 μm Mylar beamsplitter and a DLaTGS detector in the experimental frequency range with a resolution of 0.5 cm^{-1}. The film was mounted into an Oxford Optistat AC-V12w continuous flow cryostat with the film in He vapor. The experimental temperature region can be varied from 5 K to 300 K. Temperature-dependent resistance measurements were carried out on a home-built system, which is based on an Oxford Spectromag SM4000 magnetic system equipped with a series of precise electrical instruments. Here, the Sub-Femtoamp Remote SourceMeter (Keithley 6430) was used as a powerful tool for high-resistance measurements.

We adopt the density-functional theory (DFT) framework with the generalized gradient approximation (GGA) for the

primitive cell of $CuCrO_2$. The Perdew–Burke–Ernzerhof (PBE) functionals are used to address the exchange–correlation interactions.[15] To overcome the well-known underestimation of the intra-band Coulomb interactions in metal oxides including 3d electrons, some other methods with additional corrections are often adopted to obtain more accurate results. Density functional theory (DFT) with Hubbard U correction (DFT+U) has been demonstrated to be particularly efficient and effective for treating the self-interaction error in transition metal chemistry.[16] Nevertheless, the calculated results from DFT are sufficient for supporting the discussions here. Prior to performing vibration property calculations, the structure was relaxed until the forces on the atoms in the equilibrium position were smaller than 0.01 eV A^{-1} while keeping lattice parameters fixed and equal to the experimentally determined ones ($a = 6.332$ Å, $\alpha = 24.890$ Å, $u = 0.103$ Å) for the space group $R\bar{3}m$ with atoms Cu, Cr and O occupying 1a(0,0,0) 1b(0.5,0.5,0.5) and 2c(u,u,u). The energy cutoff for the plane wave basis is 400 eV. Charge densities were approximated to 1.0×10^{-6} eV per atom by the self-consistent calculations. The integration within the Brillouin zone was performed over a $11 \times 11 \times 11$ Monkhorst–Pack grid in the reciprocal space. The lattice dynamics were further assessed by the final displacement method with ultrasoft pseudopotential.

3 Results and discussion

The temperature dependent Raman scattering spectra for $CuCr_{0.97}Mg_{0.03}O_2$ are shown in Fig. 1(a). Several vibration modes can be observed from the Raman spectra. To identify the vibration modes, phonon-dispersion curves yielded by the first-principles calculations are studied, as shown in Fig. 2(a). We have analyzed the symmetry of the phonon patterns at $k_\Gamma =$
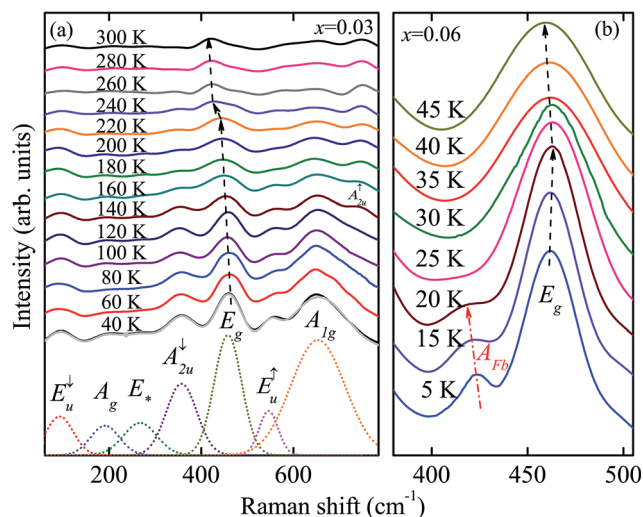

Fig. 1 (a) Raman spectra of $CuCr_{0.97}Mg_{0.03}O_2$ film, corrected by Bose−Einstein population factor, as a function of temperature from 40 K to 300 K. The Lorentzian-shaped spectral deconvolution (light blue line) at 40 K shows good consistency with the experimental result. (b) An enlarged region of Raman spectra for the $CuCr_{0.94}Mg_{0.06}O_2$ film in the extremely low temperature region from 5 K to 45 K.
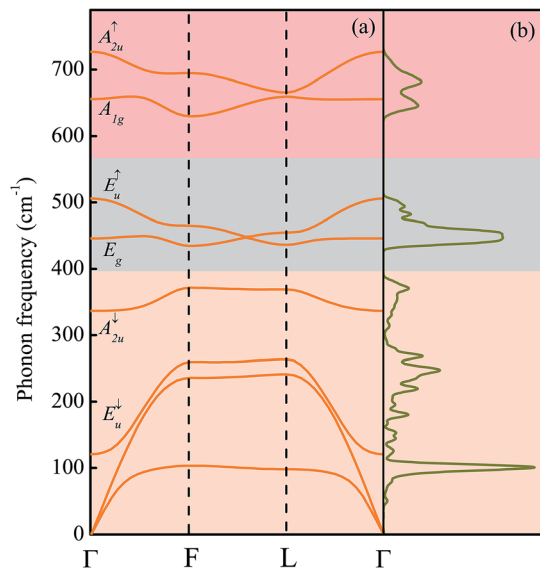
Fig. 2 (a) Calculated phonon dispersion curves for CuCrO$_2$. (b) Density of phonon states for CuCrO$_2$.

$(0,0,0)$, $k_F = (0.5, 0.5, 0)$ and $k_L = (0, 0.5, 0)$, as well as along the ΓL paths. In delafossite compounds (space group $R\bar{3}m$) such as CuGaO$_2$ and CuCrO$_2$, the primitive unit cell contains four atoms, giving rise to 12 normal modes.[17] The Cu and Cr atoms occupy D_{3d} site symmetry, while the O atom occupies C_{3v} site symmetry. Group-theoretical analysis decomposes a general mode for delafossite oxide CuMO$_2$ (M = trivalent cation) at the Brillouin zone center as:

$$\Gamma = A_{1g} + E_g + 3A_{2u} + 3E_u. \qquad (1)$$

At point Γ, the O atoms move in phase opposition while the Cu and Cr atoms are at rest for the even (g) modes. For the odd (u) modes, the O atoms vibrate in phase while the Cu and Cr atoms move with no restriction. The A modes imply movements of the Cu–O bonds along the hexagonal c-axis, whereas double
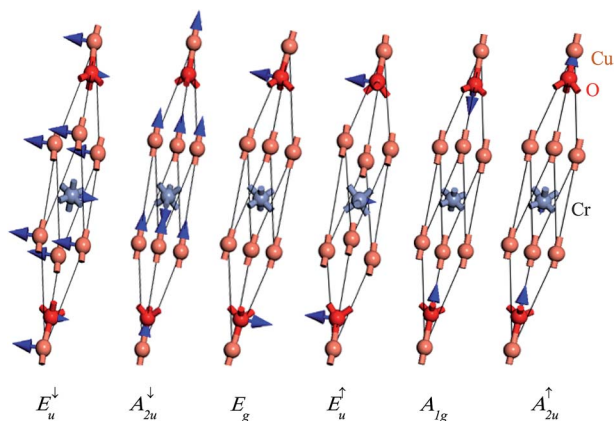

Fig. 3 Displacement pattern of each optical mode at the Brillouin-zone center.

degenerate E modes describe vibration in the triangular lattice perpendicular to the c-axis, as shown in Fig. 3. Normal vibrations are described in terms of the same irreducible representations as in the Γ point.[18] However, the symmetrization of the translation degrees of freedom requires the consideration of two Brillouin zones at the zone edge, and the criterion concerning the O vibration patterns is inverted with respect to the Γ point. From points other than Γ, group theory yields the decompositions as follows:

$$L = 4A_g + 2B_g + 2A_u + 4B_u, \qquad (2)$$

$$\Gamma L = 8A' + 4A''. \qquad (3)$$

At point L, the O atoms move in phase for the even modes and vibrate in phase opposition for the odd modes, while the metal cations are motionless for the odd modes. The A modes represent vibrations in the σ plane, and B modes represent vibrations along the twofold axis. The only symmetry operation is σ along ΓL. Vibrations take place in the σ plane for the A′ modes, whereas in the A″ modes they vibrate in the perpendicular direction.[19] It can also be observed from Fig. 2(b) that the calculated phonon dispersions along the principal directions in the Brillouin zone of CuCrO$_2$ exhibit three major bands. These bands can be distinguished directly from the observations of density of phonon states (DPS). It is suggested that the O vibration patterns dominate the highest frequency band and the Cu vibration patterns dominate the lowest frequency band, while the transition metal modes make the most contribution to the middle frequency band.[20] This indicates that the Cr^{3+}-dominated magnetoelectric properties of CuCrO$_2$ can have greater influence on the behaviour of the $E_u^\uparrow$ and E_g modes.

Focusing on the patterns at the Brillouin zone center, one of the three E_u odd modes represents an acoustic mode, while the other two modes are infrared-active. The same is the case for the A_u modes. Even modes denoted with the "g" subscript are Raman-active. As can be seen from the phonon-dispersion curves, the calculated normal modes arranged in ascending frequencies are $E_u^\downarrow$ (120 cm^{-1}), $A_{2u}^\downarrow$ (336 cm^{-1}), E_g (446 cm^{-1}), $E_u^\uparrow$ (505 cm^{-1}), A_{1g} (655 cm^{-1}), and $A_{2u}^\uparrow$ (726 cm^{-1}). Both Raman and infrared-active modes can be observed from the Raman spectra. The infrared-active modes of the $E_u^\downarrow$ peak, $A_{2u}^\downarrow$ peak and $E_u^\uparrow$ peak can be found around 100 cm^{-1}, 360 cm^{-1} and 546 cm^{-1}, respectively. The appearance of the mode at about 193 cm^{-1} at the lowest temperature is assigned to the A_g mode. Note that the weak mode around 270 cm^{-1}, labeled as E_*, is supposed to be from the sapphire substrate.[14,21] The modes around 460 cm^{-1} and 650 cm^{-1} at 40 K can be assigned to the E_g and A_{1g} modes, respectively. The observed mode centers are quite in accordance with the theoretical ones. Meanwhile, the temperature-dependent evolution of the modes shows a rather complicated behavior. With increasing temperature, most modes become weaker and gradually fade away. However, the $A_{2u}^\uparrow$ peak above 740 cm^{-1} shows a growing trend with increasing temperature. This is because the $A_{2u}^\uparrow$ mode considerably polarizes the unit cell and thus creates an important TO/LO splitting.

The competition between the splitting modes indicates the anisotropy in the $CuCrO_2$ systems. The splitting effect may also appear in other optical modes, but is relatively small.[21] The original A_{1g} mode and some other modes merge into one broadened peak with decreasing temperature. This is ascribed to the enhanced Cr 3d–O 2p–Cu 3d hybridization with decreasing temperature.[22] On the other hand, the peak center of the E_g mode shows a generally declining trend below 300 K, as can be seen in Fig. 1(a). The blueshift of the peak center with decreasing temperature is usually attributed to the electron–phonon interaction and lattice thermal expansion. Interestingly, there is a sudden drop at the temperature above a cross point (T_{cross}) around 220 K, which has not been reported in previous Raman studies on pure $CuCrO_2$. This phenomenon can also be found for $CuCr_{0.94}Mg_{0.06}O_2$ and $CuCr_{0.91}Mg_{0.09}O_2$ at a temperature of around 200 K, as can be seen in Fig. 4(b). This is because there is a transition of the carrier transport mechanism from a thermal activation behavior to a 3D variable range hopping (VRH) one at a specific temperature T_{cross} for impurity-doped $CuCrO_2$.[23,24] The transition behavior has been discovered previously from a complex relationship between the resistance and temperature, whereas the decline of the E_g mode peak center offers more clear evidence for the transition.[25] The results also provide further evidence for the 3D HTAF model in the low temperature region mentioned above.

Fig. 1(b) shows the Raman spectra for $CuCr_{0.94}Mg_{0.06}O_2$ in the lower temperature region from 5 K to 45 K. The detailed temperature dependence of the E_g mode center from the best fitting results is shown in Fig. 4(a). The peak center of the E_g mode for the film doped with $x = 0.03$ gradually diminishes with decreasing temperature to 5 K. This is in accordance with the abnormal band gap redshift trend with decreasing temperature at relatively low temperatures reported previously.[26] The anomalous behavior in the low temperature region is related to the strong Cr 3d–O 2p–Cu 3d interaction. The occupied Cr 3d states interact covalently with the neighboring O states and enhance the p–d hybridization. Meanwhile, it was
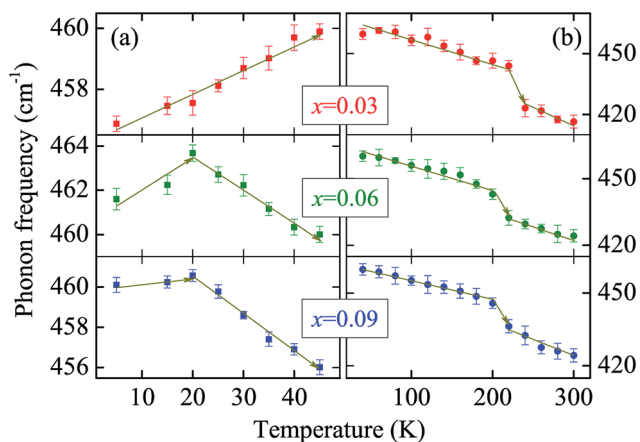
also manifested that the interaction can be weakened by heavy Mg doping due to the stronger disturbance of local spin fluctuation at Cr sites.[14,25] This point can be manifested by the normal blueshift of the E_g peak center with decreasing temperature for the heavily doped $CuCr_{1-x}Mg_xO_2$ ($x = 0.06$ and 0.09) just below 45 K. However, it is obvious that the peak center of the E_g mode for the heavily doped $CuCr_{1-x}Mg_xO_2$ again shows a redshift trend with decreasing temperature when the temperature drops below 20 K. It can also be found from Fig. 1(b) that an additional mode just between $A_{2u}^{\downarrow}$ and E_g also appears with decreasing temperature below 20 K. According to previous studies, there are two successive magnetic transitions for $CuCrO_2$ near the temperature around 24 K.[10] The turning temperature is well in accordance with the temperature at which an additional mode around 420 cm^{-1} appears, indicating that they may share the same origin as the magnetic transitions.

The infrared spectra for $CuCr_{1-x}Mg_xO_2$ have also been studied at different temperatures as a complement to the Raman scattering experiments. The enlarged region of the $E_u^{\uparrow}$ mode is shown in Fig. 5(a). The frequency of $E_u^{\uparrow}$ here is lower than that observed from the Raman spectra. This is also ascribed to the TO/LO splitting. Further, this splitting is more obvious in the $E_u^{\uparrow}$ and $A_{2u}^{\uparrow}$ modes, indicating that the unit cell is more polarized in these modes than in the $E_u^{\downarrow}$ and $A_{2u}^{\downarrow}$ modes.[27] The peak around 546 cm^{-1} in the Raman spectra can be assigned to the $E_u^{\uparrow}$ (LO) mode while the relatively smaller one from the infrared spectra here is suggested to be the $E_u^{\uparrow}$ (TO) mode. The peak center of the $E_u^{\uparrow}$ mode firstly shifts to a higher
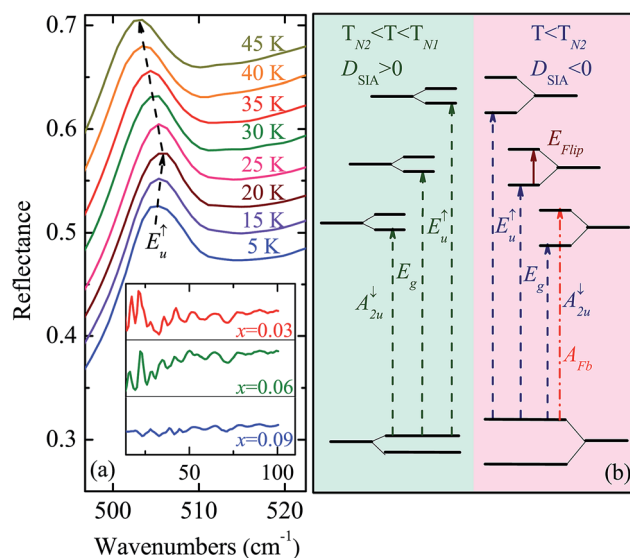

Fig. 4 (a) Temperature dependence of the E_g mode center in the temperature region of 40–300 K for $CuCr_{1-x}Mg_xO_2$ films. (b) Temperature dependence of the E_g mode center in the temperature region of 5–45 K for $CuCr_{1-x}Mg_xO_2$ films.

Fig. 5 (a) The $E_u^{\uparrow}$ mode center from infrared spectroscopy for the $CuCr_{0.94}Mg_{0.06}O_2$ film in the low temperature region. The inset shows the vibration for the far-infrared spectroscopy in the region below 100 cm^{-1} at 5 K. Note that the vibration state has been normalized with the same scale for each sample. (b) An energy level spacing diagram for the $CuCr_{1-x}Mg_xO_2$ films was applied to explain the experimental results. The transition shown by the solid arrow-line represents one of the spin-flip energies. The forbidden transition shown by the dash-dot arrow-line corresponds to the absorption peak labeled as A_{Fb} in Fig. 1(b).

frequency with decreasing temperature, which is mainly caused by the lattice thermal shrinkage. However, when the temperature drops below 20 K, the peak center of the $E_u^\uparrow$ mode shifts to a lower frequency with decreasing temperature. This trend is similar to the temperature dependence of the E_g mode from the Raman spectra, but with a more obvious feature. The abnormal temperature dependence of the E_g and $E_u^\uparrow$ modes is ascribed to the influence on the electronic excitations from the crystal field-induced spin flip. It has been manifested that far-infrared spectroscopy offers a unique possibility to directly obtain the energy-level splitting between the low-lying crystal field split states without any further assumptions of the symmetry or the coupling to the neighboring spins.[28] The far-infrared spectroscopy for films doped with $x = 0.03$ and 0.06 show several sharp peaks below 100 cm^{-1} at 5 K, as can be seen from the inset of Fig. 5(a). These relatively strong vibrations are supposed to be the features of the splitting energies from spin flips, corresponding to the excitation energy labeled by E_{Flip} in Fig. 5(b).[29] Nevertheless, the vibration for the $CuCr_{0.91}Mg_{0.09}O_2$ film below 100 cm^{-1} is very similar to that of sapphire substrates, which is much weaker than that of samples with $x = 0.03$ and 0.06. This is because heavily doped Mg can disturb the spin fluctuation around Cr^{3+} and further attenuate the spin flip effect.

To explore the intrinsic correlation between these abnormal phenomena and magnetic phase transitions, the magnetoresistance (MR) of $CuCr_{1-x}Mg_xO_2$ at different temperatures is studied. The shoulder on the temperature-dependent trend of the resistivity under a magnetic field directly induces the magnetic phase transitions, as shown in Fig. 6(a). Negative MR can be observed in the whole variable range-hopping region (not fully shown here) due to enhanced localization around Cr sites.[30] What catches our eyes is that the temperature-dependent MR shows a sharp vibration across magnetic transition temperature regions, as shown in the inset of Fig. 6(a). The absolute values of MR for $CuCr_{0.94}Mg_{0.06}O_2$ under different magnetic fields were plotted in Fig. 6(b). The MR shows an

obvious peak below a the temperature T_{N1} around 24.7 K, with an additional valley at a strong magnetic field above 5 Tesla. The MR vibration for $CuCr_{1-x}Mg_xO_2$ has been found before but hasn't been discussed in detail so far.[23] The frustrated magnetic structure of $CuCrO_2$ is a TLA, with three spins forming 120° angles to neighboring spins.[11] The magnetic properties for Heisenberg-type antiferromagnetism with single-ion anisotropy (SIA) can be described on the basis of the following Hamiltonian:

$$\hat{H} = J \sum_{i,j}^{chains} S_i S_j + J' \sum_{k,l}^{planes} S_k S_l$$
$$+ D_{SIA} \sum_i \left(S_i^z\right)^2 - g\mu_B H \sum_i S_i, \qquad (4)$$

where S is a spin of the magnetic ion, J is the exchange integral along the c axis of the crystal, and D_{SIA} represents the single-ion anisotropy constant.[31] On the other hand, the Cr^{3+} levels for $CuCr_{1-x}Mg_xO_2$ in an octahedral environment contain two low lying spin-allowed (but dipole forbidden) excitations derived from the $^4T_{2g}$ and $^4T_{1g}$ states of the excited $t_{2g}^2 e_g^1$ configuration.[32] The octahedral crystal-field distortion splits the ground states by an energy associated with D_{SIA}, although the single-ion anisotropy splitting is normally quite small.[33] According to the Tanabe–Sugano diagram, the increasing octahedral field with decreasing temperature drives the ground state shifts from the $^4T_{2g}$ to $^4T_{1g}$, and finally turns D_{SIA} negative at a certain temperature T_{N2} around 23 K.[34,35] On the other hand, the typical 120° spin structure of TLA is determined by the sign of D_{SIA}. For a triangular antiferromagnet with positive D_{SIA}, the spin spiral plane for the 120° structure is parallel to the triangular lattice plane (TLP), while the 120° structure shows an out-of-plane characteristic for those with negative D_{SIA}.[36]

The enhancement of negative MR around the magnetic transition temperature is usually due to a strong spin-charge coupling effect.[23] When the temperature decreases below T_{N1}, the structure of $CuCr_{1-x}Mg_xO_2$ first comes into a collinear antiferromagnetic (CAF) phase ($T_{N2} < T < T_{N1}$), whose state can usually be achieved in a triangular lattice including additional hopping processes beyond the nearest neighbours.[37] Positive MR can increase in a CAF state due to the influence on spin fluctuations from localized magnetic moments.[30] The strong competition between the localization and CAF configuration leads to the sharp vibration of negative MR just below T_{N1}. When the temperature further decreases below T_{N1}, the out-of-plane structure enhances the spin-charge coupling, leading to the strong enhancement of the negative MR. Note that T_{N1} and T_{N2} are slightly different from the previously reported results due to Mg doping influences.

Let's move our attention back to the optical mode behavior. An energy level spacing diagram for the $CuCr_{1-x}Mg_xO_2$ films was applied to explain the experimental results, as shown in Fig. 5(b). The phonon energy is related to the transition between the ground state and the excited energy level taking the spin flip at low temperatures into account. In the CAF temperature region, the internal spin fluctuation is weak due to the in-plane structure. With decreasing temperature below T_{N2}, the growing
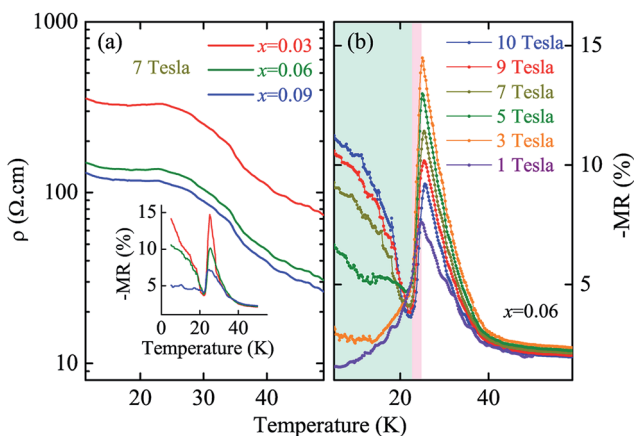

Fig. 6 (a) Temperature dependence of the resistivity for $CuCr_{1-x}Mg_xO_2$ films under a magnetic field of 7 Tesla. The inset shows the absolute value of MR for $CuCr_{1-x}Mg_xO_2$ films under 7 Tesla. (b) The absolute value of MR for the $x = 0.06$ compound under different magnetic fields.

spin fluctuation due to the out-of-plane structure increases the splitting energy from the spin flip. The energy gap between the ground state and the excited state decreases. Thus, both the E_g and $E_u^{\uparrow}$ mode centers show a red-shift trend with decreasing temperature below T_{N2}. The additional mode just between $A_{2u}^{\downarrow}$ and E_g should also be associated with the spin flip energy, corresponding to the forbidden transition energy labeled as A_{Fb} in Fig. 5(b). In order to create a conventionally dipole-forbidden excitation, one needs to overcome the single-ion anisotropy energy. In the temperature region above T_{N2}, the spin flip is quite small, while in the temperature region below T_{N2}, the single-ion anisotropy energy increases. Thus, the additional mode shows a blueshift with decreasing temperature, as can be seen in Fig. 1(b). The negative MR near zero magnetic field below T_{N2} is very weak [as can be seen from Fig. 6(b)], manifesting that the spin fluctuation is relatively strong. The out-of-plane structure further enhances the spin fluctuation below T_{N2}, which again increases the Cr 3d–O 2p–Cu 3d hybridization. This moves the mode center to lower energy with decreasing temperature for heavily doped films at zero magnetic field.

4 Conclusions

In conclusion, phonon modes from Raman and infrared spectra for Mg-doped $CuCrO_2$ films have been investigated, combined with first-principles calculations. The transition of the carrier transport mechanism from a thermal activation to a 3D variable range hopping one results in a decline in the temperature dependence of the E_g mode peak center. The abnormal redshift of the E_g mode center with decreasing temperature below 40 K for the lightly doped film ($x = 0.03$) is due to p–d hybridization. The E_u and E_g mode centers for heavily doped films show a normal blueshift just below 40 K owing to stronger disturbance of the local spin fluctuation at Cr sites from Mg doping. The increasing spin-charge coupling caused by the out-of-plane structure below T_{N2} not only results in the enhancement of negative MR at strong magnetic field, but also brings back the abnormal redshift below T_{N2} for heavily doped films.

Acknowledgements

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